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## Classe Scienze Fisiche Matematiche Naturali RENDICONTI

Elio Cannillo, Giuseppe Rossi, Luciano Ungaretti The crystal structure of macdonaldite

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# Mineralogia. - The crystal structure of macdonaldite ${ }^{(*)}$. Nota di Elio Cannillo, Giuseppe Rossi e Luciano Ungaretti, presentata ${ }^{\text {**) }}$ dal Socio G. Carobbi. 

RiASSunto. - La macdonaldite è un silicato la cui formula chimica, tratta dalla letteratura, è: $\mathrm{BaCa}_{4} \mathrm{Si}_{15} \mathrm{O}_{35}$. II $\mathrm{H}_{2} \mathrm{O}$; quattro di queste unità stechiometriche sono contenute nella cella elementare. Il gruppo spaziale è Cmcm ; le costanti reticolari, rideterminate per il presente lavoro, sono le seguenti: $a=14,08 \mathrm{I}, b=\mathrm{I} 3$, $109, c=23,560 \pm \mathrm{o}, \mathrm{oOI} \AA$.

La struttura è stata determinata attraverso l'esame della sintesi di Patterson tridimensionale e di successive sintesi di Fourier tridimensionali. Il raffinamento è stato eseguito col metodo dei minimi quadrati. Il fattore di discordanza finale per i riflessi osservati è o.083. I poliedri di coordinazione del calcio sono ottaedri formati da atomi di ossigeno e da molecole d'acqua; questi ottaedri, mettendo in comune uno spigolo, formano catene parallele ad $a$. Tali catene sono collegate fra loro attraverso legami idrogeno e formano strati paralleli a (OOI). Tra due strati ottaedrici è posto un doppio strato tetraedrico costituito da anelli di quattro e otto tetraedri $\mathrm{SiO}_{4}$. Gli strati tetraedrici sono del tipo apophyllite «idealizzato». L'impalcatura tetraedrica è attraversata da un doppio sistema di canali: uno parallelo a $b$ con diametro utile di $2,1 \AA$, l'altro parallelo ad $a$ con diametro utile di circa $3,4 \AA$. All'interno dei canali si trovano gli atomi di bario coordinati da sei ossigeni dei tetraedri e da quattro molecole d'acqua. Sono pure presenti nei canali molecole d'acqua non legate ai cationi.

Viene discussa l'analogia strutturale della macdonaldite e con le zeoliti fibrose e con i minerali rhodesite e delhayelite.

La formula chimica è modificata, sulla base dell'analisi strutturale, nel modo seguente: $\mathrm{BaCa}_{4} \mathrm{H}_{2} \mathrm{Si}_{16} \mathrm{O}_{38} \cdot(8+x) \mathrm{H}_{2} \mathrm{O}$ dove $x$ è uguale a 2,4 per il cristallo usato nella presente ricerca.

## Introduction.

Macdolnaldite is one of the seven barium minerals recently found in Eastern Fresno County, California, and described by Alfors et al. [r]. The following data are quoted from the Authors cited above:

$$
\begin{array}{ll}
\text { lattice parameters } & a=14.06 \pm 0.01 \AA \\
& b=13.08 \pm 0.01 \AA \\
& c=23.52 \pm 0.02 \AA \\
\text { space group } & \mathrm{Cmcm} \\
\text { cell content } & 4\left[\mathrm{BaCa}_{4} \mathrm{Si}_{15} \mathrm{O}_{35} \cdot \text { I I } \mathrm{H}_{2} \mathrm{O}\right] .
\end{array}
$$

The mineral occurs in crystals elongated following [ioo].
These cleavages are present: $\{001\}$ perfect, $\{010\}$ good, $\{100\}$ poor or a fracture.

In respect to the paper of Alfors et al. [I], who choose the axes setting corresponding to the convention used in Crystal Data, the reference axes have been interchanged in order to obtain an orientation consistent with the standard setting of the International Tables of X-rays Crystallography.
(*) This work was performed in the Sezione di Pavia del Centro Nazionale di Cristallografia del C.N.R., Istituto di Mineralogia dell'Università, Via Bassi 4, 27100 Pavia.
(**) Nella seduta del i9 novembre 1968.

## Experimental.

The sample used for the crystal structure analysis was a prismatic fragment elongated following $a$. The side dimensions of the rectangular section of the sample were: 0.0200 cm (side parallel to $b$ ), 0.0096 cm (side parallel to $c$ ).

The lattice parameters were re-determinated and the values obtained are:

$$
\begin{aligned}
& a=14.08 \mathrm{I} \pm 0.00 \mathrm{I} \AA \\
& b=\mathrm{I} 3.109 \pm 0.00 \mathrm{I} \AA \\
& c=23.560 \pm 0.00 \mathrm{I} \AA .
\end{aligned}
$$

Integrated equi-inclination Weissenberg photographs of the $h k l$ reflexions ( $h$ from o to io) were taken with nickel filtered $\mathrm{CuK} \alpha$ radiation, using the multiple film technique. A total of 1948 reflexions, out of the 2626 present in the $\mathrm{CuK} \alpha$ limiting sphere (about $75 \%$ ) were inspected; 1428 of them were measured photometrically, 520 were too faint to be suitably measured or did not give any blackening on the films.

The intensities were corrected for the Lorentz-polarization and absorption factors and for the incipient but incomplete $\alpha_{1}-\alpha_{2}$ spot doubling. The absorption factors were obtained through an exact integration over the whole diffracting volume -- considered as continuous - of the crystal; the formulas given by Cannillo and Mazzi [2] were used. The linear absorption coefficient is $\mu=\mathrm{I} 6 \mathrm{I} .8 \mathrm{~cm}^{-1}$ and the transmission factor ranges from I to 4 on a relative scale.

The correction for the $\alpha_{1}-\alpha_{2}$ splitting effect was applied taking into account the integration technique which complicates the splitting effect for its diagonal direction with respect to the sides of the film.

## Crystal structure analysis.

A first inspection of the Weissenberg pictures permitted to observe that the average intensity of the reflexions with $h=2 n+$ I was noticeably lower than that of the reflexions with $h=2 n$. Furthermore, among the reflexions of the latter set, those with $h=4 n$ have the strongest intensities. The pseudo-symmetry inferred from these observations was taken into account in the course of the structure analysis.

From a three-dimensional Patterson synthesis it was possible to locate all the atoms of $\mathrm{Ba}, \mathrm{Ca}$ and Si . A three-dimensional Fourier synthesis, computed giving to the Fo's the phases of the heavy atoms contributions, permitted to find the positions of the remaining atoms.

The structural model obtained in this way is self consistent only with the assumption that there are 16 Si atoms for each Ba atom in the chemical
formula and no alternative structure is possible with the number of Si and O atoms of the chemical formula given by Alfors et al. The chemical formula will be fully discussed later.

The arrangement of the atoms in the crystal structure of macdonaldite gives a full account of the intensity differences in the sets of reflexions mentioned above. All the atoms but barium are arranged in such a way that the glide plane at $x=\mathrm{I} / 4$ in the Cmcm space group becomes a mirror plane. Thus the contributions of these atoms to the structure factors of the reflexions with $h=2 n+i$ are near to zero. The strong average intensity of the reflexions with $h=4 n$ is explained by the fact that many atoms are crowded on planes at $x=0, x=\mathrm{I} / 4, x=\mathrm{I} / 2$ etc.

The discrepancy factor at this stage was 0.28 for all the observed reflexions.

## Refinement.

Five cycles of least-squares calculation carried out on the observed reflexions, using the ORFLS program by Busing, Martin and Levy, with isotropic temperature factors for all the atoms, lowered the R factor to o.II.

At this stage it was observed that two oxygen atoms, O (II) and O (i7), considered as water molecules, had temperature factors exceedingly high. This fact, together with the difficulties encountered in their location, suggested the following procedure. The multipliers of all the oxygen atoms were allowed to vary in the course of a cycle of the least-squares refinement. The result was a strong diminution of the multipliers of O (II) and O (I7) while the remaining oxygens showed only slight variations in both directions. Thus some doubt on the presence of those atoms arose.

In order to solve the question a structure factors calculation, without O (II) and O (I7) was carried out and the relative $F_{0}$ Fourier synthesis was examined. This one appeared very disturbed by series termination effects and gave no conclusive information. In order to eliminate such perturbations a further Fourier synthesis was computed using $F O \cdot \exp B \frac{\sin ^{2} \vartheta}{\lambda^{2}}$ as coefficient ( $\mathrm{B}=3.0$ ) instead of Fo's. Most of the disturbances disappeared while, the maxima corresponding to O (II) and $\mathrm{O}(\mathrm{I} 7)$ remained and their heights were those expected on the basis of the multipliers obtained from the least-squares refinement. An usual $\Delta F$ synthesis confirmed these observations. All these data were interpreted in terms of an incomplete occupation of the positions the two water molecules involved.

In the difference synthesis there was also a sure evidence that Ba and Ca had anisotropic thermal factors. The three successive least-squares cycles were computed varying the multipliers of O (II) and O ( I 7 ) and using anisotropic temperature factors for Ba and Ca atoms. It has not been thought worthwhile to extend the anisotropic treatment to all the atoms because of the uncertainity on the number of water molecules and because the scale factor was not unique.

## Table I.

Final atomic parameters and their standard deviations (in parentheses).
The letter $w$ labels those oxygen atoms which are considered as belonging to water molecules. The asterisk (*) is used to distinguish the equivalent isotropic temperature factors, after Hamilton [18], of those atoms which were treated anisotropically in the refinement.

| Atoms | $x / a$ | $y / b$ | $z / c$ | B |
| :---: | :---: | :---: | :---: | :---: |
| Ba . | o | 0.2054(I) | 0.2500 | 0.70* ${ }^{\text {(4) }}$ |
| $\mathrm{Ca}(\mathrm{I})$ | o | $0.2522(3)$ | -0.0022(I) | - $.55^{*}(8)$ |
| $\mathrm{Ca}(2)$ | 0.2500 | 0.2500 | $\bigcirc$ | 0.40* ${ }^{\text {(8) }}$ |
| $\mathrm{Si}(\mathrm{I})$ | 0.1098(3) | 0.1473(2) | o. 1153 (I) | 0.17 (4) |
| Si(2) | 0.3895(3) | 0.1406(2) | 0.1170(1) | 0.24 (4) |
| Si(3) | 0.2524(3) | $0.2695(2)$ | o.1835(I) | 0.08 (4) |
| Si(4) | 0.2522(3) | $0.4642(2)$ | 0.1025(I) | O. 10 (4) |
| $\mathrm{O}(\mathrm{I})$ | 0.1212(6) | $0.2073(6)$ | $0.0582(3)$ | 1.14 (12) |
| $\mathrm{O}(2)$ | 0.1542(6) | $0.2098(6)$ | $0.1701(3)$ | I. 28 (12) |
| $\mathrm{O}(3)$ | $0.2536(7)$ | 0.4234(5) | $0.0386(3)$ | 1.12 (12) |
| $\mathrm{O}(4)$ | 0.1568(8) | $0.0352(6)$ | 0.1149(3) | 1.76 (15) |
| $\mathrm{O}(5)$ | 0.257 I (7) | $0.3699(5)$ | 0.1476(3) | 1.21 (12) |
| $\mathrm{O}(6)$ | 0.3751(6) | 0.1998(6) | $0.0599(3)$ | 1.17 (12) |
| $\mathrm{O}(7)$ | -. 3464 (8) | 0.0251 (7) | -. 1159 (4) | 2.37 (17) |
| $\mathrm{O}(8)$ | - 3419 (6) | -.1974(6) | 0.1714(3) | 1. 20 (12) |
| $\mathrm{O}(9) w$. | o | 0.4217 (15) | $0.0404(8)$ | 5-36 (43) |
| $\mathrm{O}(\mathrm{IO}) w$ | o | $0.3830(15)$ | 0.1825(8) | 5.52 (45) |
| $\mathrm{O}(\mathrm{I}$ ) $) z^{\prime}$ | 0. 5000 | $0.3823(23)$ | -.1682(12) | 4.63 (98) |
| $\mathrm{O}(12) w$ | 0.5000 | $0.4188(17)$ | $0.0407(9)$ | 6.43 (51) |
| $\mathrm{O}(\mathrm{I} 3) w$. | o.105 I(I2) | 0.0227 (10) | 0.2500 | 2.74 (24) |
| $\mathrm{O}(\mathrm{I} 4)$. | $\bigcirc$ | 0.1319(7) | -.1348(4) | 0.64 (16) |
| $\mathrm{O}(\mathrm{I} 5)$ | 0.2491 (9) | $0.3002(8)$ | 0.2500 | 1.22 (17) |
| $\mathrm{O}(16)$ | 0. 5000 | 0.1239(8) | 0.1339(4) | 1.07 (18) |
| $\mathrm{O}(17) w . .$. | -.3574(26) | 0.0112(26) | 0.2500 | 6.30 (1.2I) |

The final R factor is 0.083 for the observed reflexions and o.II6 for all the reflexions.

The anomalous dispersion correction for barium was carried out with the method proposed by Patterson [3]; $\Delta f^{\prime}$ and $\Delta f^{\prime \prime}$ are given by Cromer [4].

The secondary extinction effect appeared to be nearly negligible and no correction was applied.

Table II. Analysis of the anisotropic thermal parameters.
(Root mean square thermal vibration along the ellipsoid axes $(\AA)$, magnitudes of the principal axes $\left(\AA^{2}\right)$ and angles ( ${ }^{\circ}$ ) between the crystallographic axes and the principal axes of the vibration ellipsoid).

| Atom | r.m.s. | B | $\alpha$ | $\beta$ | $\gamma$ |
| :---: | :---: | :---: | :---: | :---: | :---: |
| Ba . . . . . . . . . | 0.09 | 0.72 | 0 | 90 | 90 |
|  | 0.12 | I. 15 | 90 | - | 90 |
|  | 0.05 | 0.22 | 90 | 90 | O |
| $\mathrm{Ca}(\mathrm{I})$. . . . . . . . | 0.06 | 0.30 | 90 | I 18 | 28 |
|  | 0.12 | I. 16 | 90 | 28 | 62 |
|  | 0.05 | 0.21 | - | 90 | 90 |
| $\mathrm{Ca}(2)$. . . . . . . . | 0.05 | 0.23 | I I | 90 | 79 |
|  | O.II | 0.97 | 93 | I6 | 74 |
|  | O.OI | O.OI | 100 | 106 | 19 |

Final atomic coordinates and thermal parameters with their standard deviations are given in Table I. The analysis of the anisotropic thermal parameters of $\mathrm{Ba}, \mathrm{Ca}(\mathrm{I})$ and $\mathrm{Ca}(2)$ is reported in Table II; of course their significance is lessened by the fact that not all the atoms were treated anisotropically and that the structure factors were scaled level by level. In Table III the final observed and calculated structure factors are compared.

## Discussion.

The interatomic distances and bond angles are presented in Table IV as well as their standard deviations.

Calcium. Both calcium atoms in the asymmetric unit have a six-fold coordination. $\mathrm{Ca}(\mathrm{I})$ is linked to four oxygens belonging to the $\mathrm{Si}-$ tetrahedra and to two water molecules. $\mathrm{Ca}(2)$ which lies on an inversion center, is surrounded by six oxygens of the Si-tetrahedra. The coordination polyhedra of both Ca atoms could be considered as squares bipyramids rather than octahedra because there are four oxygens forming a square around Ca at distances ranging from 2.29 to $2.35 \AA$, and two oxygens, at opposite sides of the square, with longer $\mathrm{Ca}-\mathrm{O}$ distances (from 2.42 to $2.45 \AA$ ).

TAble III.
Structure factors of macdonaldite.
Reflexions marked with an asterisk were unobservably weak; in this case Fo derives from $0.5 \mathrm{I}_{\mathrm{min}}$


Table III (continued).

|  | 1 10F0 | F |  | 1 10Fo |  |  |  |  |  |  |  |  |  |  |  | 1 10Fo |  |  |  |  |  |  | 107c |  | 1 10Fo |  | $k 11070$ | 107e |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  | -867 | 115 | 1392 | 1428 | 1* |  | -234 | 13 2* | 2* 440 | 324 |  | 114.468 |  | 105 | 51024 | -1135 |  | 1925 | -992 |  | 731 | -623 | $\bigcirc 10$ | - 2006 | -1849 | 1796 | 1604 |
| 10 | ${ }_{5} 1033$ | 1192 | 16* | 5* 440 | 258 | 72 | 21313 | 1408 | $13^{3} 3$ | 3* 436 | -648 |  | 124203 | c939 | 106 | 61824 | -2001 | 122 | 2 Les | -290 |  |  | 1045 | 012 | 21053 | -1018 | 612144 | -643 |
|  | 6*490 | -586 | 117 | 955 | -973 | 3* | 3* 401 | 241 | $13{ }^{4 *}$ | ${ }^{4 *} 434$ | -637 |  | 13*477 | 495 | 107 | 946 | 945 | 123 | $\mathrm{ol}^{\circ} \mathrm{P}$ | 1102 | $7: 3$ | 3630 | 553 | - 14 | $4{ }^{645}$ | 532 | 132639 |  |
| 10 | 7* 497 | 448 | 1 18* | 8* 436 | -251 | 74 | 4 7月9 | -A46 | 13 5* | 5* 430 | 566 |  | 14** 40 | 12 | 108 | 1737 | -1226 | ${ }_{1}{ }^{24 *}$ | 6* 293 | 89 | 716 | 61028 | -1043 | $\bigcirc 16$ | 61905 | 2104 | 141264 | -1215 |
| 10 | 81025 | 1124 | 119 | 1351 | 1420 | 5* | 5* 427 | -50s | 13 6* | 6*424 | 304 |  | 152059 | -1918 | 109 | 1790 | -1954 | 125 | 648 |  | 717 | 7678 | 617 | - 18 | 8754 | -740 | 15721 | 141 |
| 10 | 9* 509 | 489 | 204 | 204 417 | 191 | 76 | 61197 | 1195 | 13 7* | 7*440 | - 529 |  | 161827 | 1804 | 1010 | 103592 | -3839 | 126 | 313 | -372 | 718 | 8945 | 947 | - 20 | 2* 310 | -2 | 6161339 | 1306 |
| 10 | 10* 512 | 307 | 21 | 809 | -799 | 77 | 7047 | 887 | 13 sm | \$* 430 | -499 |  | 17760 | -331 | 10 11* |  | 267 | 3.0 |  |  | 719 | 9* 297 |  | - 22 | 22347 | -2605 | 6171004 | -829 |
|  | 11* 512 | -134 | $122 \times$ | 2* 382 | -345 | 78 | 81229 | -1288 | 13 | 9832 | 860 |  | 18* 406 | 331 | 1012 | 12* 449 | 592 | 1 | 11031 | 988 | 720 | - 509 | -543 | 024 | 41840 | 2200 | 6 18* 326 | 147 |
| 10 | 12904 | 912 | 123 | 959 | 91 | 7 9* |  | -626 | $13{ }^{10}{ }^{\text {a }}$ | 417 | 1 |  | 19* 384 | -477 | 10 | 89 | 123 | 32 | 973 |  | 721 | 1*259 | -362 | - | O 1454 | -1370 | 192624 | 608 |
|  | 131244 | -1194 | $124 *$ | 4*348 | 4 A |  | 0 1298 | 1190 | 1311 | 1711 | -741 |  | 202349 | 2340 | 10 14* | 4**21 | -329 | ${ }^{3} 3$ | 3399 | -299 | 722 | 233 | 609 | 1 | 12360 | -264 | 620661 | -602 |
| 10 | $1625 n 0$ | -2416 | 125 | 612 | -680 | $11+$ | 1+ 525 | -8 | $13{ }^{12 *}$ | 2* 364 | -340 |  | 212084 | 2020 | 10 15* | 5* 387 | 236 | 4 | 4540 | -553 | 723 | 3340 | 412 | 2 | 21002 | -1036 | 211002 | 1008 |
|  | 151022 | ${ }^{1} 11$ | 26* | 6* 298 | -274 | 12 | 21112 | -1049 | 1313 | 3640 | 728 |  | 221898 | 1996 | 10 16* | 16* 360 | -294 | 35 | 684 | 733 | c | c* 373 | 324 | 3 | 31368 | -1462 | 22* 215 | 2 |
| 10 | 161131 | 1036 | ${ }_{1} 27$ | 876 | 1092 | 13* | 3* 551 | -505 | 13 14* | 4* 291 | 336 |  | 23904 | -960 | 10 17* | 17*334 | -316 | 36 | 6652 | 561 | 91 | 992 | 1092 |  | 4780 | 142 | 861 | 31 |
|  | 17 547 | -432 | 0 | 01156 | -1076 | 14 | 41243 | 1088 | 1315 | 5494 | -551 |  | 24* 295 | 87 | 10 | 182308 | -2503 | 37 | 71205 | 122 | 92 | 698 | -685 | 5 | 5* 293 | 205 | 1*372 |  |
|  | 19** 372 | 439 | 31 | 1010 | 1019 | 15* | 5* 525 | 191 |  |  |  |  | $25 \quad 454$ | 390 | 10 19* | 19* 269 | -105 | 38 | 8 774 | -707 |  |  | -568 | 6 | 63313 | 3588 | 832 | 911 |
| 10 | 10607 | 495 | 32 | 1140 | 1196 | 16 | 6971 | -897 |  |  |  |  | $26 \quad 584$ | -659 | * | 11 | 134 | 39 | 1085 | 1071 |  | 799 | 857 | 7 | 72895 | 2905 | 32329 | -2555 |
|  |  |  |  | 31143 | -1289 | 17* | 7* 462 | -355 |  | 8 k 1 |  |  | O 1177 | -1203 | 12 o | 1487 | 1569 | 310 | 1091 | 948 |  |  | 421 | 8 | 83673 | -35s5 | 41675 | 1675 |
| 10 | 21* 283 | -210 | 34 | 1028 | 1186 | 18 | 981 | 817 |  |  |  | 6 | 907 | 85 | 12 | 423 | 13 | 311 | 1871 | -720 | 96 | 611 | -622 | 9 | 92019 | -1814 | 51409 | 495 |
|  | 22721 | -961 | 35 | 65s | 764 | 19* | 9* 405 | 274 | 02 | 21024 | 992 |  | 21348 | -1557 | 122 | 21600 | 1680 | 312 | -897 | -798 | 97 | 688 | -690 | 210 | 0* 403 | -213 | 6* 398 | -43 |
| 12 | - 434 | -790 | 36 | 1287 | 1443 | 20 | - 793 | -859 | - | 4582 | 147 | 6 | 32413 | 2463 | 123 | 1428 | 1569 | ${ }^{3} 13 \pm$ | 3* 389 | 391 | 98 | 8* 400 | 147 | 211 | 11591 | -1213 | 71371 | 487 |
| 12 | 1*484 | -373 | 37 | 832 | 19 | 21 | 1650 | -623 | - | 6651 | -376 | 6 | 42769 | -2935 | 124 | 1389 | 1322 |  | 4897 | 845 | 99 | 829 | 773 | 212 | 2905 | 836 | 8769 | 786 |
| 12 | 21062 | 1153 | 38 | 1295 | -1405 | 22 | 2998 | 1093 | 0 R | \% 3661 | 3672 | 6 | 52565 | -2740 | 12 | 777 | -653 | 315 | 5968 | -889 | 910 | 956 | -898 | 213 | 3* 376 | -224 | * 416 | 407 |
| 12 | 31298 | -1326 | 39 | 704 | 746 | 230 | 3* 310 | 260 | 10 | - 813 | 519 | 6 | 63020 | -3209 | 12 | 2410 | 2644 | 316 | 635 | -644 | 11 | 1599 | -551 | 214 | $1{ }^{1} 1054$ | 1049 | 10769 | 739 |
| 12 | ${ }_{4} 1031$ | -1212 | 310 | 1061 | 1032 | 24 | 4698 | -835 | 012 | 21548 | 1302 | 6 | 71352 | -1080 | 12 | 2210 | -2448 | 17* | 7* 348 | 317 | 912 | 2643 | 607 | 15 | 51955 | 2012 | 111823 | -1812 |
| 12 | 5* 492 | -31 | ${ }^{3} 11$ | 127 | 1193 | 0* | 0* 449 | 437 | $\bigcirc 14$ | ${ }_{4} 2761$ | -2976 | 6 | 1156 | 98 | 128 | 426 | 45 | 18 | 879 | 62 | 91 | 946 | 914 | 16 | 6* 350 | 289 | 12144 | 1306 |
| 12 | 6*490 | -367 | 12 | 2951 | -962 | 1* | 1* 452 | 648 | 16 | 63897 | 4490 |  | 1415 | -1426 | 129 | 638 | 581 | 319 | 775 | -766 | 914 | 14*376 | -352 | 217 | 7642 | -524 | 13* 424 | 220 |
| 12 | 72458 | 2654 | 13 | 877 | A26 | 2* | 2* 456 | -600 | 18 | 81532 | 1436 | 10 | 101275 | -1062 | 1210 | tc 1509 | 1506 | 320 | 750 | -718 | 915 | 5730 | -713 | 218 | 8*340 | -277 | 14* 390 | -282 |
| 12 | R ${ }^{4} 490$ | -217 | 14 | 41206 | 1137 | 93 | 3946 | $-1011$ |  | 2026 | 2719 |  | 113973 | 3993 | 1211 | 111406 | 1508 | 321 | 1671 | 645 | 916 | 6529 | 47 | 219 | 9* 334 | 243 | 15528 | -482 |
| 12 | 915 | -849 | 15 | 51039 | -1054 | 4* | 465 | 717 | 22* | 2* 351 | -84 |  | 122182 | 1704 | 1212 | 12* 387 | 146 | 322 | 276 | 723 | 917 | 7741 | 72 | 220 | - 1806 | -1883 | 16* 334 | -450 |
| 12 | 101487 | 1505 | 16 | 783 | -737 | 95 | 027 | 1135 | - 24 | 4551 | 664 |  | 13929 | 662 | 1213 | 13* 354 | -351 | 323 | 891 | -934 | 918 | 8829 | -820 | 221 | 11071 | -1094 | 172105 | 2151 |
| 12 | 111025 | -987 | 17 | 7655 | 726 | 6* | 484 | -625 |  | 26* 101 | 492 | 614 | 141482 | -1119 | 1214 | 141787 | 1795 | 324 | 4465 | -454 | 919 | 9348 | -343 | 222 | 21933 | 2069 | 181175 | 1286 |
| 12 | 12** 462 | -631 | 18 | 8 -874 | 864 | 97 | B52 | -882 | 20 | 3977 | 416 |  | 15780 | -780 | 1215 | 15* 283 | -211 | 325 | 5343 | 419 | 92 | - 555 | 540 | 223 | 3845 | 840 | 191324 | 1347 |
| 12 | 13* 460 | -112 | 19 | 9673 | 695 | 8* | ${ }^{8 *} 503$ | 49 |  | 1123 | 1002 |  | 161042 | -886 | 1216 | 16* 228 | 18 |  | 01754 | 1786 |  | 0* 373 | 93 | 2218 | 2333 | -416 | 20758 | -860 |
| 12 | 14* 397 | 559 | 20 | - 859 | -890 | 9 9 | 987 | 1059 | 22 | 286 | 303 | 17 | 171289 | 1293 | 140 | 662 | 639 | 1* | 1*279 |  | 11 | 667 | -702 | 0 | ¢ 3509 | 3808 | 10 0* 380 | 85 |
| 12 | $15 \quad 599$ | 540 | 210 | 1*395 | 420 | 10 | - 931 | -855 | 3 | 31001 | -708 |  | 19* 392 | -121 | 141 | 1009 | 1184 | 2 | 1400 | -1507 |  | ${ }^{2 *} 373$ | 101 | 1 | 11632 | 1541 | 1011648 | 1957 |
| 12 | 161414 | - 1372 | ${ }^{3} 22$ | 2809 | 830 |  | 1* 525 | -610 | 24 | 43036 | -3324 | d | 191474 | 1322 | 142 | 21268 | -1278 | 53 | 3492 | 505 | 3 | 31219 | 1331 | 2 | 2* 281 |  | 102606 | -654 |
| 12 | 17987 | 912 | 23 | 3796 | -870 | 9 12* | 2* 528 | 736 | $5 \times$ | 5+ 292 | -27 |  | 20695 | -730 | 143 | 3* 381 | -308 |  | 890 | 992 |  | 4* 386 | -120 | 3 | 31165 | -1128 | 10 3* 382 | -333 |
|  | 18 689 | 861 | 24 | 713 | -795 | 13 | 3823 | 874 |  | 1671 | -1678 |  | 211132 | 1063 | 14 | 674 | 731 |  | 5* 324 | -407 |  | 51203 | -1293 | 4 | 43198 | 3655 | 41481 | 1555 |
| 14 | 0*433 | 134 | 25 | 500 | 612 |  | 4* 525 | -623 | 27 | 2818 | 3054 |  | 222588 | -2720 | 14 | 607 | 468 | 56 | 1377 | -1431 |  | 6* 389 | 187 | 45 | 748 | 738 | 10 5* 403 | 716 |
| 14 | ${ }^{1 *} 433$ | 393 | 26 | 665 | 792 |  | 5* 481 | -682 |  | 1283 | -972 |  | 23862 | -959 | $14{ }^{6}$ | 938 | -1077 |  | 355 | 283 |  | 7* 386 | 347 |  | 6* 340 |  |  | -872 |
| 14 | ${ }^{2+431}$ | -554 | 27 | 7577 | -724 | 16* | 6* 440 | 235 | , | 1632 | 1180 |  | $24 \quad 683$ | -656 | 147 | 740 | 678 | 58 | 1275 | 1283 | 118 | 8* 380 | 300 | 7 | 71729 | -1722 | 107769 | 748 |
| 14 | 3* 427 | -306 |  | - 1318 | 1246 | 17 | 7807 | 854 | ${ }^{2} 10 *$ | 10* 435 | 228 |  | - 2007 | 1860 | $14{ }^{8}$ | 8 8 323 | -17 |  | 9569 | - 511 |  | 864 | $-863$ |  |  | -584 | $\begin{array}{lll}10 & 8 * 406\end{array}$ | 234 |
| 14 | 4880 | 746 | 51 | 1540 | -444 | 18* | 8* 389 | -415 | ${ }^{2} 11$ | 1679 | -439 | 8 | 1* 401 | -305 | 14 | 9* 278 | -150 | 510 | - 957 | -968 |  | 0* 373 | 220 | 9 | 91589 | 1453 | 10 9* 400 | 285 |
| 14 | 5* 422 | 447 | 2 | 21419 | -1390 |  | 9 585 | -592 | 212 | 22736 | 2732 | 8 | ${ }^{2} 888$ | 807 |  | 101418 | -1954 | 511 | 1772 |  |  |  | 1174 |  | O 2954 | 2822 | $1010 * 400$ | -548 |
|  |  | -775 | 3* | 3* 339 | 10 |  | - 339 | 469 | 213 | 31115 | 971 | 8 | 3+412 | -238 |  |  |  | 512 | 2912 | 961 | 1112 | 2* 353 | -111 | 4 11* | 1* 443 | -322 | 10 11* 398 | -320 |
| 14 | 8* 408 | 73 | 55 | 293 | 1266 | 21 | 518 | 747 | 214 | 14 668 | -669 | 8 | 4.1268 | -1157 |  |  |  |  | 3* 433 | 358 | 1113 | 3928 | -925 | 412 | 21670 | 1468 | 10121574 | 1466 |
| 14 | 91609 | 1558 | 6 | 61158 | -1109 | 11. | 0*481 | -205 | 216 | 64540 | 4893 | 8 | 63207 | 3249 |  |  |  | $515 *$ | 5*389 | 14 |  | 5533 | 558 | 4 | ${ }_{4}{ }^{2} 4.412$ |  |  | -926 |
| 14 | 10*389 | -149 | 74 | 7* 405 | 127 | 111 | 863 | -879 | 217 | 71217 | -1248 | 8 | 71337 | 1490 |  |  | 24 | ${ }_{5} 16$ | 61419 | 1424 | 1116 | 6* 234 |  | 415 | 52189 |  | 14* 151104 |  |
|  | 111080 | 007 | 8 | 81258 | 1223 | $11{ }^{2+}$ | 2* 481 | 84 | 218 | 18* 398 | -245 | 8 | 81600 | 1601 | 1 | 928 | -1184 | 5 17* | 7* 348 | -330 |  | 7876 | -1063 | 416 | 61374 | -1344 | 10 16* 256 | -7 |
| 14 | 121412 | 1616 |  | 9* 446 | 115 | 113 | 3958 | 1009 | 219 | 91460 | -1570 |  | $9 * 474$ | 217 |  | 531 | -568 | 519 | 9 8.56 | -872 | 130 |  | -44 | 417 | 72277 | 2256 | 1017611 | -7 |
|  |  |  |  | O 1450 | -1410 | $11{ }^{11} 4$ |  |  | 220 | 2098 | -921 | 81 | 102711 | 2641 | 3 | 31232 | 1446 | 520 | O 736 |  | 131 |  | 330 | 418 | 81989 | -1910 | 12 0* 374 | 51 |
|  | 7 k 1 |  |  | 511 | 1*493 | 199 | 115 | 995 | -921 | 221 | 211019 | -978 |  | 111931 | -1804 | 4 | 4423 | 466 |  | 1*282 |  | 132 |  | 399 | 419 | 9* 324 | -56 | 12121448. | 1541 |
|  |  |  |  |  | 21179 | 1021 |  |  | 109 | 222 | 2262 | -604 |  | 12* 498 | 202 | 15 | 1509 | -1567 | 522 | 2946 | -997 | 13 | 759 | -758 | 420 | 2 1258 | 1303 | 122579 | 6 |
|  |  |  |  | S 513 | 3* 541 | ${ }^{-361}$ | $11{ }_{11} 7$ | 7*487 | 88 | 223 2 24 1 | 23 1669 | 1750 |  | 13792 | 560 |  | 619 | -534 | 523 | 344 | 446 | 134 | 4755 | -741 | 421 | 1838 | 763 | $12 \quad 31259$ | 1011 |
|  | $\begin{aligned} & 1 \\ & 2 \\ & 2 \\ & \hline \end{aligned} 5827 .$ | ${ }_{-1}^{-1900}$ | ¢ 14 | $\begin{array}{ll}4 & 1318 \\ 519\end{array}$ | -1108 | ${ }_{11}^{11}{ }^{8}$ | 8* 487 |  | 224 | 21827 | -1594 | 81 | 14.1688 | 1454 | 7 | 71547 | 1479 | 524 | 704 | 856 | 13 | 5929 | 1011 | 422 | 2* 273 | -129 | 1241417 | -1516 |
| 1 | 31393 | 1556 | 516 | 6974 | 916 | 1110 | 10* 481 | ${ }_{-} 824$ |  | (1) 290 | -437 | 815 | 151093 | 932 | 8 | $550$ | 520 | - | - 825 | -856 | 136 |  |  |  |  | 131 | $12 \quad 51509$ | 1549 |
| $1$ | 4607 | 474 | 517 | 7* 468 |  | 1111 | 1889 | 856 | 0 | ${ }_{0} 1814$ | -474 164 |  | 161505 | ${ }_{-535}^{1408}$ |  |  | 1049 -636 |  |  |  |  |  |  |  |  |  | 12 ${ }_{12} 6 \times 364$ | 362 |
| 1 | 51145 | -1075 | 518 | 81208 | -1197 | $11_{12}{ }^{\text {+ }}$ | ${ }^{\text {2* }} 468$ |  | 1 | 12170 | 2094 |  | 181661 | -1486 | 111 | 111280 | ${ }_{1} 1304$ |  |  | 478 |  |  |  |  |  |  |  | 317 507 |
| 1 | ${ }^{6 *} 380$ | $-343$ |  | 9* 430 |  | 1113 | 31092 | -1050 | 2 | 718 | 600 | ${ }_{8} 1$ | 191536 | -1614 | ${ }_{1}^{1} 12$ | 12706 | 1364 626 | 74 | 3* 4 1123 | -1185 |  | - 489 | 498 | 6 | 2* 335 | -517 |  | 2427 |
| $\begin{aligned} & 1 \\ & 1 \end{aligned}$ | 71360 $8 * 415$ | 1222 303 |  | 20 957 | 1009 | ${ }^{11} 1^{14}$ | 14* 411 | 157 | 3 | 2388 | 2314 | 8 | 20.655 | -428 | 113 | 131024 | -1068 |  | 5* 366 | -152 |  |  | 4 | 63 | ${ }^{2 \times 2} 234$ | 2301 | 12101380 | 1364 |
| $\begin{aligned} & 1 \\ & 1 \end{aligned}$ | $8 * 415$ <br> 9 <br> 1288 | -1265 | 22 | 21* 380 | -972 |  | 15 611 | 610 -35 |  |  | ${ }_{2} 21216$ |  |  | 1850 1590 | $1 \begin{array}{ll}1 & 14 \\ 1 & 15\end{array}$ | $\begin{array}{lll}14 & 597 \\ 15 \\ 15\end{array}$ |  |  |  | 1010 325 |  |  |  |  |  | -4003 | 12111434 | 99 |
| $1$ | 10*417 | -314 | ${ }_{5} 23$ | 3**34 | 342 | 1117 | $17 \quad 798$ | -813 | 5 | 812 |  | ${ }^{8} 10$ | - 21.1312 | ${ }_{-1835}^{159}$ |  | 15* ${ }^{1276}$ |  |  |  | ${ }_{-521}$ |  |  |  |  | 52714 $6 * 388$ |  |  | - |
|  | 111407 | 1401 | 24 | 24919 | 982 | 1118 | 18*300 | -78 | 7 | 72195 | -1923 | 10 | 1*437 | 474 | 1.17 | 17743 | -786 |  | 9915 | -796 | 0 |  | -305 | 7 | 7* 398 |  |  |  |
|  | 12*427 | -1266 | 525 526 | 25* 298 |  | 1119 | 19626 | 721 | 48 | 8738 | -438 | 10 | 21830 | -1891 | 118 | $18 * 334$ | -363 | 710 | 10 1049 | 990 | - | 41396 | 1276 | 8 | 8* 408 | $\begin{array}{r} 305 \\ 99 \end{array}$ |  |  |
|  | 131224 | 1266 |  |  |  | 130 | 0*442 | -274 | 49 | 91491 | 1395 | 10 | 3* 442 | -92 |  |  | 1130 |  | 1* 431 | 368 | - 6 | 62452 | -2466 | 69 | 736 | 678 |  |  |
|  | 436 | -158 | - | - 1253 | -1364 |  | 1768 | 97 | 10 | 101988 | 1805 | 10 | 41896 | 1982 | 120 | 20631 | 637 | 712 | 2815 | -720 | O 8 | 83180 | 3377 | 610 | 0* 430 | 431 |  |  |

Barium. Barium lies on a mm equipoint and so its coordination polyhedron is rather regular. This cation is surrounded by six oxygens lying at the corners of a slightly folded hexagon and by four water molecules. Two out of them are located above the hexagon, on a mirror plane, and two below, on another mirror plane normal to the former. The barium-oxygen system has $a_{1}$ compact structure that is comparable with a close packed arrangement of spheres.

The $\mathrm{Ba}-\mathrm{O}$ distances range from 2.82 to $2.87 \AA$; these values are very near to the sum of the ionic radii of the involved atoms ( $2.86 \AA$ ).

Silicon. Four silicon atoms occur in the asymmetric unit of macdonaldite. One silicon shares all the oxygens with other tetrahedra; the remaining three have one oxygen unshared. The bridging $\mathrm{Si}-\mathrm{O}$ bonds of the latter three silicon atoms are appreciably different from the non-bridging bonds. The mean lengths of the bridging bonds are: $\mathrm{Si}(\mathrm{I})-\mathrm{O}$ I. $629 \AA, \mathrm{Si}(2)-\mathrm{O}$ I. $626 \AA$, $\mathrm{Si}(4)-\mathrm{O} \mathrm{I}^{2} 625 \AA$; the lengths of the non-bridging bonds are: $\mathrm{Si}(\mathrm{I})-\mathrm{O}(\mathrm{I})$ r. $567 \AA, \operatorname{Si}(2)-O(6) \quad 1.567 \AA, \operatorname{Si}(4)-O(3)$ 1.597 $\AA . \quad$ Such differences are consistent with the $d-p \pi$-bonding hypothesis suggested by Cruickshank [5].

Table IV.
Interatomic distances $(\AA)$ and angles $\left({ }^{\circ}\right)$ and their standard deviations (in parentheses).
An asterisk is used to distinguish equivalent atoms. The distances preceded by the sign ' occur twice; those preceded by " occur four times.

| Atoms | Bond lengths | Atoms | Bond angles. |
| :---: | :---: | :---: | :---: |
| $\mathrm{Si}(\mathrm{I})-\mathrm{O}(\mathrm{I})$ | 1.567 (7) | $\mathrm{O}(\mathrm{I})-\mathrm{Si}(\mathrm{I})-\mathrm{O}(2)$ |  |
| - $\mathrm{O}(2)$ | 1.651 (7) | $\mathrm{O}(\mathrm{I})-\mathrm{Si}(\mathrm{I})-\mathrm{O}(4)$ | $114^{\circ} 15^{\prime} \quad\left(28^{\prime}\right)$ |
| - $\mathrm{O}(4)$ | I.610 (8) | $\mathrm{O}(\mathrm{I})-\mathrm{Si}(\mathrm{I})-\mathrm{O}(\mathrm{I} 4)$ | ${ }_{11} 3^{\circ} 43^{\prime} \quad\left(28^{\prime}\right)$ |
| -O(I4) | I. 625 (8) | $\mathrm{O}(2)-\mathrm{Si}(\mathrm{I})-\mathrm{O}(4)$ | ${ }_{107}{ }^{\circ} 34^{\prime} \quad\left(28^{\prime}\right)$ |
| $\mathrm{Si}(2)-\mathrm{O}(6)$ | 1.567 (7) | $\mathrm{O}(2)-\mathrm{Si}(\mathrm{I})-\mathrm{O}(14)$ | 101 $^{\circ} 33^{\prime}$ (28) |
| $-\mathrm{O}(7)$ | 1.631 (9) | $\mathrm{O}(4)-\mathrm{Si}(\mathrm{I})-\mathrm{O}(\mathrm{I} 4)$ | ${ }_{1060}{ }^{\circ} 3^{\prime}\left(28^{\prime}\right)$ |
| -O(8) | 1.625 (7) | $\mathrm{O}(6)-\mathrm{Si}(2)-\mathrm{O}(7)$ | $113^{\circ} 25^{\prime \prime} \quad\left(28^{\prime}\right)$ |
| -O(16) | 1.620 (8) | $\mathrm{O}(6)-\mathrm{Si}(2)-\mathrm{O}(8)$ | $113^{\circ}{ }^{2} 2^{\prime}$ (28) |
| $\mathrm{Si}(3)-\mathrm{O}(2)$ | 1.620 (9) | $\mathrm{O}(6)-\mathrm{Si}(2)-\mathrm{O}(16)$ | ${ }_{11} 3^{\circ} 42^{\prime} \quad\left(28^{\prime}\right)$ |
| $-\mathrm{O}(5)$ | 1. 565 (7) | $\mathrm{O}(7)-\mathrm{Si}(2)-\mathrm{O}(8)$ | $106{ }^{\circ} 32^{\prime}$ (28) |
| -O(8) | I.601 (9) | $\mathrm{O}(7)-\mathrm{Si}(2)-\mathrm{O}(16)$ | 103 ${ }^{\circ} 37^{\prime \prime}$ (28) |
| -O(15) | 1.617 (6) | $\mathrm{O}(8)-\mathrm{Si}(2)-\mathrm{O}(16)$ | $105^{\circ} 21^{\prime}$ (28) |
| $\mathrm{Si}(4)-\mathrm{O}(3)$ | I. 597 (6) | $\mathrm{O}(2)-\mathrm{Si}(3)-\mathrm{O}(8)$ | ${ }_{110^{\circ}} 33^{\prime}$ (28) |
| $-\mathrm{O}(4)^{*}$ | I. 62 I (9) | $\mathrm{O}(2)-\mathrm{Si}(3)-\mathrm{O}(5)$ | ${ }_{109}{ }^{\circ} 41^{\prime} \quad\left(28^{\prime}\right)$ |
| $-\mathrm{O}(5)$ | 1.633 (7) | $\mathrm{O}(2)-\mathrm{Si}(3)-\mathrm{O}(15)$ | ${ }_{1060} 37^{\prime}\left(28^{\prime}\right)$ |
| - $\mathrm{O}(7)^{*}$ | 1. 622 (II) | $\mathrm{O}(8)-\mathrm{Si}(3)-\mathrm{O}(5)$ | $\mathrm{III}^{0} 26^{\prime}$ (28) |
| $\mathrm{Ca}(\mathrm{I})-\mathrm{O}(\mathrm{I})$ | 2.299 (8) | $\mathrm{O}(8)-\mathrm{Si}(3)-\mathrm{O}\left(\mathrm{I}_{5}\right)$ | $\begin{array}{llll}110^{\circ} & 3^{\prime} & \left(288^{\prime}\right)\end{array}$ |
| -O(6)* | 2.308 (8) | $\mathrm{O}(5)-\mathrm{Si}(3)-\mathrm{O}(\mathrm{I} 5)$ | $1080{ }^{2} 2^{\prime}\left(28^{\prime}\right)$ |
| -O(9)w | 2.437 (20) | $\mathrm{O}(5)-\mathrm{Si}(4)-\mathrm{O}(7)^{*}$ | ${ }^{1060} 45^{\prime} \quad\left(28^{\prime}\right)$ |
| - O (12) 2 u | 2.417 (22) | $\mathrm{O}(5)-\mathrm{Si}(4)-\mathrm{O}(4)^{*}$ | $106^{\circ} 4^{\prime \prime} \quad\left(28^{\prime}\right)$ |
| $\mathrm{Ca}(2)-\mathrm{O}(\mathrm{I})$ | 2.340 (7) | $\mathrm{O}(5)-\mathrm{Si}(4)-\mathrm{O}(3)$ | $\begin{array}{llll}111^{\circ} & 4^{\prime} & (28)^{\prime}\end{array}$ |
| $-\mathrm{O}(3)$ | ' 2.447 (6) | $\mathrm{O}(7)^{*}-\mathrm{Si}(4)-\mathrm{O}(4)^{*}$ | $\begin{array}{lll}111^{\circ} & 2^{\prime} & \left(288^{\prime}\right.\end{array}$ |
| -O(6) | '2.350 (7) | $\mathrm{O}(7)^{*}-\mathrm{Si}(4)-\mathrm{O}(3)$ | 1110 ${ }^{\circ} 29^{\prime}$ (28) |
| $\mathrm{Ba}-\mathrm{O}(2)$ | " 2.874 (7) | $\mathrm{O}(4)^{*}-\mathrm{Si}(4)-\mathrm{O}(3)$ | $110^{\circ} 15^{\prime}$ (28) |
| -O(14) | ' 2.880 (9) | $\mathrm{Si}(\mathrm{I})-\mathrm{O}(\mathrm{I} 4)-\mathrm{Si}(\mathrm{I})^{*}$ | $144^{\circ} 4^{\prime} \quad\left(56^{\prime}\right)$ |
| -O(ro) w | ${ }^{\prime} 2.820$ (20) | $\mathrm{Si}(\mathrm{I})-\mathrm{O}(2)-\mathrm{Si}(3)$ | ${ }^{1} 35^{\circ} 41^{\prime} \quad\left(36^{\prime}\right)$ |
| $-\mathrm{O}(\mathrm{I} 3) w$ | '2.814 (14) | $\mathrm{Si}(3)-\mathrm{O}(\mathrm{r} 5)-\mathrm{Si}(3)^{*}$ | ${ }_{151}{ }^{\circ}$ (56) |
| $\mathrm{O}(3)-\mathrm{O}(3)^{*}$ | 2.71 (2) | $\mathrm{Si}(3)-\mathrm{O}(5)-\mathrm{Si}(4)$ | ${ }_{17} 70^{\circ} 24^{\prime} \quad\left(36^{\prime}\right)$ |
| $\mathrm{O}(9) w-\mathrm{O}(9) w^{*}$. | 2.80 (6) | $\mathrm{Si}(3)-\mathrm{O}(8)-\mathrm{Si}(2)$ | ${ }^{1} 37^{\circ} 24^{\prime}$ ( $36^{\prime}$ ) |
| $\mathrm{O}(\mathrm{I} 2) w-\mathrm{O}(\mathrm{I} 2) w^{*}$ | 2.87 (6) | $\mathrm{Si}(2)-\mathrm{O}(16)-\mathrm{Si}(2)^{*}$ | $147^{\circ} 27^{\prime}$ (56') |
| $\mathrm{O}(\mathrm{II}) w-\mathrm{O}(\mathrm{I} 3) w$ | 3.05 (3) | $\mathrm{Si}(4)-\mathrm{O}(7)^{*}-\mathrm{Si}(2)^{*}$ | ${ }^{140^{\circ}} 53^{\prime} \quad\left(36^{\prime}\right)$ |
| $\mathrm{O}(17) w-\mathrm{O}(10) w^{*}$ | 3.06 (3) | Si 4 )- $\mathrm{O}(4)^{*}-\mathrm{Si}(\mathrm{I})^{*}$ | $148^{\circ} 28^{\prime}\left(36^{\prime}\right)$ |
| $\mathrm{O}(8)-\mathrm{O}(17) w$ | 3.07 (3) |  |  |
| $\mathrm{O}(2)-\mathrm{O}(\mathrm{I})$ w | 3.16 (2) |  |  |
| $\mathrm{O}(2)-\mathrm{O}(13) w$ | $3 \cdot 17$ (1) |  |  |

The tetrahedron around $\mathrm{Si}(3)$ needs a more detailed discussion. $\mathrm{Si}(3)$ shares all its oxygens with other tetrahedra and the average $\operatorname{Si}(3)-\mathrm{O}$ distance is 1.60 I $\AA$. It must be pointed out that the $\mathrm{Si}(3)-\mathrm{O}(5)$ distance is the shortest of all the $\mathrm{Si}-\mathrm{O}$ distances of macdonaldite. Pant and Cruickshank [6] suggest that large $\mathrm{Si}-\mathrm{O}-\mathrm{Si}$ angles " cause the bridging bonds to gain strength at the expense of peripheral bonds ". Actually the $\mathrm{Si}(4)-\mathrm{O}(5)-\mathrm{Si}(3)$ angle is $170^{\circ}$ and this fact could explain the short $\mathrm{Si}(3)-\mathrm{O}(5)$ distance. On the other hand the second bond distance, $\mathrm{Si}(4)-\mathrm{O}(5)$, involved in the cited angle, is $1.633 \AA$. One could expect, on the basis of the cited large angle, a shorter length for the $\operatorname{Si}(4)-\mathrm{O}(5)$ bridging bond, but one must take into account the different $\pi$-orders existing in the $\mathrm{Si}(4)$-tetrahedron. However the average of the two $\mathrm{Si}-\mathrm{O}(5)$ distance is $\mathrm{I} .60 \AA$ that is a value which could be expected for a $\mathrm{Si}-\mathrm{O}-\mathrm{Si}$ angle of $170^{\circ}$. Fig. I shows a plot of the $\mathrm{Si}-\mathrm{O}-\mathrm{Si}$ angles of some recently studied silicates versus the average of the two $\mathrm{Si}-\mathrm{O}$


Fig. I. - Plot of the average of the two $\mathrm{Si}-\mathrm{O}$ distances involved in one $\mathrm{Si}-\mathrm{O}-\mathrm{Si}$ angle versus the $\mathrm{Si}-\mathrm{O}-\mathrm{Si}$ angle for some recently studied silicates.
bridging bonds lengths involved. One can observe a rather regular shortening of the $\overline{\mathrm{Si}-\mathrm{O}}$ distances (from a value of about $\mathrm{I} .63 \AA$ ) with the increasing, of the angles (from a value of about $137^{\circ}$ ).

The $\mathrm{O}-\mathrm{Si}-\mathrm{O}$ angles range from $\mathrm{Ior}^{\circ}$ to $\mathrm{II} 4^{\circ}$. By plotting the mean lengths of any pair of bonds forming an angle at Si , against the $\mathrm{O} \cdots \mathrm{O}$ distances, the results of this work fit the corresponding curve given by McDonald and Cruickshank [7].

Balance of electrostatic valences. The balance of electrostatic valences computed in the usual way, assigning a bond strength of $\mathrm{I} . \mathrm{Oo}$ to tetrahedral $\mathrm{Si}-\mathrm{O}$ bonds, $\mathrm{I} / 3$ to $\mathrm{Ca}-\mathrm{O}$ bonds and $\mathrm{I} / 5$ to $\mathrm{Ba}-\mathrm{O}$ links, give some unacceptable results: $\mathrm{O}(2)$ and $\mathrm{O}(\mathrm{I} 4)$ are "overbonded" (2.20) while $\mathrm{O}(\mathrm{I})$, $\mathrm{O}(6)$ and $\mathrm{O}(3)$ are " underbonded " (i.66, i. 66 and I. 33 respectively). The valences get balanced if one accept the suggestion of Zachariasen [9] that "the observed bond lengths uniquely determine the bond strengths". In
this way $\mathrm{O}(\mathrm{I})$ and $\mathrm{O}(6)$, which form with silicon non-bridging bonds of $\mathrm{I} .57 \AA$, would have a bond strength greater than $\mathrm{I} .00 ; \mathrm{O}(2)$ and $\mathrm{O}(14)$ which form with silicon bridging bonds of $\mathrm{I} .62-\mathrm{I} .63 \AA$, would have a bond strength less than i.oo. As Pant [8] observes " ...the two theories ( $d-p \pi$-bonding theory and the method of balancing of valences) for the $\mathrm{Si}-\mathrm{O}$ bonds are not exclusive. $\pi$-bonding in $\mathrm{Si}-\mathrm{O}$ bonds may be part of the mechanism whereby valency balance is achieved '".


Fig. 2. - Perspective view of the chains of Ca -octahedra and of a layer of tetrahedra. The rings of four and eight tetrahedra are shown; the layer is doubled by a mirror plane parallel to $x y$.

For the case of $\mathrm{O}(3)$ it is necessary to spend some more words. In fact, for this atom the valences balance is not achieved even with the application of the cited method; $\mathrm{O}(3)$ is linked to $\mathrm{Si}(4)$ with a non-bridging bond of I. $597 \AA$ and to $\mathrm{Ca}(2)$; in such a way the positive charge reaching this oxygen is about I. 4 that is far from 2.0. On the other hand, if this atom is considered to be a hydroxyl group, it would be "overbonded" ( $1.4+$ I.0). Furthermore, the chemical formula would have an excess of positive charges.' The $\mathrm{O}(3)-\mathrm{O}\left(3^{\prime}\right)$ distance of $2.70 \AA$ is the shortest distance between oxygens not belonging to the same coordination polyhedron and surely represents a hydrogen bond,
but it seems too long (Zachariasen [9]) for the hypothesis of a hydrogen atom occurring midway between $\mathrm{O}(3)$ and $\mathrm{O}\left(3^{\prime}\right)$. Thus the possibility of a statistical distribution of one hydrogen between the two equivalent $O(3)$ seems to be the more convenient one. This assumption permits to get balanced the valences of $\mathrm{O}(3)$ and the sum of positive and negative charges in the chemical formula.

Description of the structure. The crystal structure of macdonaldite is characterized by the presence of double layers of Si-tetrahedra connected by layers of Ca -octahedra ${ }^{(1)}$.


Fig. 3. - Perspective view of the structure following $b$. The ring of tetrahedra normal to $b$ is shown. The small black circles represent water molecules occurring in the channels and those linked to barium. The number in square parentheses are the $y$ coordinates. The chains of octahedra occur at $z=0$ and $\mathrm{I} / 2$.

The Ca-octahedra, sharing two opposite edges, form chains parallel to $a$. The chains are linked to one another with hydrogen bonds and give rise to sheets parallel to (OOI) and occurring at $z=0$ and $z=1 / 2$.

The tetrahedral framework is located between two of such octahedral sheets. The basic unit of the silicate framework is a ring of four tetrahedra. The rings are linked together to form a corrugated layer of tetrahedra parallel to (OOI) (See fig. 2). The presence of the mirror plane at $z=1 / 4$
(I) The coordination polyhedra of calcium have been described as square bipyramids, but in the course of this description they will be designated more simply as octahedra.
makes the layer doubled. Each single layer is very similar to the " idealized" apophyllite layer which is derived from the condensation of wollastonite chains through the xonotlite ribbon. The doubling of the layers produces two sets of channels built up by eight-membered rings of tetrahedra: one set is parallel to $a$ with a free diameter of $3.4 \AA$ and the other is parallel to $b$ with a free diameter of 2.I $\AA$ (figs. 3 and 4). The barium atoms and the water molecules not linked to calcium occur in these channels.


Fig. 4. - Perspective view of the structure showing the ring of tetrahedra that forms channels parallel to $a$.

The fact that the thermogravimetric curve of macdonaldite resembles those of mesolite and scolecite [I] and the presence of such channels, suggested the possibility of some relationship between fibrous zeolites and macdonaldite. In effect some structural arguments give strength to this hypothesis. Fig. 5 (left) shows the structural feature characteristic of natrolite [io] and of all the fibrous zeolites: it is a ring of four tetrahedra (two Si and two Al ) with alternate vertices pointing upwards and downwards; a fifth Si -tetrahedon links the upper vertices and, repeated $6.6 \AA$ below, the lower ones to form a sort of string parallel to $c$. The vertices labelled i to 4 connect together neighbouring strings thus forming a three-dimensional tetrahedral framework. The openings among neighbouring chains give place to a double system of channels (with free diameter of 2.08 and $2.6 \AA$ ) formed by eight-membered rings of tetrahedra. Fig. 5 (right) shows the four Si-tetrahedra and one of
the two Ca -octahedra contained in the asymmetric unit of macdonaldite. This group could be derived from that showed in the left part of the figure by substituting in this one the tetrahedron labelled 4 with an octahedron. The figs. 3 and 4 permit to see how this structural feature of macdonaldite (four tetrahedra and one octahedron) is repeated in the structure and the role played by the octahedra of the second Ca atom.

The water molecules. Water is present in the coordination polyhedra of barium and calcium and, as "free" or zeolitic water, in the channels of the tetrahedral framework. $\mathrm{Ca}(\mathrm{I})$ is linked to two water molecules, $\mathrm{O}(9)$ and O (I2) lying on a mirror plane. Each of these water molecules form a hydrogen bond which together with that occurring between $\mathrm{O}(3)$ and $\mathrm{O}\left(3^{\prime}\right)$ contributes to hold together the chains of Ca -octahedra.


Fig. .5. - Left: the structural feature characteristic of fibrous zeolites; dark tetrahedra are $\mathrm{AlO}_{4}$ groups. Right: the structural feature of macdonaldite built up by four Si-tetrahedra and by one Ca-octahedra; the Ca atom involved is that labelled $\mathrm{Ca}(\mathrm{I})$.

The water molecules occurring in the channels and not linked to the cations are labelled $\mathrm{O}(\mathrm{II})$ and $\mathrm{O}\left(\mathrm{I}_{7}\right)$. They occupy two positions, on mirror planes, related by a pseudo-mirror plane to those of O (io) and $\mathrm{O}(\mathrm{I} 3)$. The occupancy is $64 \%$ for $\mathrm{O}(\mathrm{II})$ and $56 \%$ for $\mathrm{O}(\mathrm{I} 7$ ); the unit cell contains 9.6 of such water molecules. They are linked very weakly to the water molecules belonging to the coordination polyhedron of barium and to some oxygens of the Si-tetrahedra.

It must be pointed out that there is room in the channels for other atoms (cations or water) and the number of free water molecules could vary without disturbing the structure.

Chemical formula. On the basis of all the considerations made in the preceding discussions and of the fact that the number of $\mathrm{H}_{2} \mathrm{O}$ linked to calcium and barium could not be varied without important changes in the
structure, the chemical formula of macdonaldite can be written as follows:

$$
4\left[\mathrm{BaCa}_{4} \mathrm{H}_{2} \mathrm{Si}_{16} \mathrm{O}_{38} \cdot(8+x) \mathrm{H}_{2} \mathrm{O}\right]
$$

where $x$ is the number of free water molecules; in the case of the present work $x=2.4$.

This formula is somewhat different from that given by Alfors et al. [I]. However the chemical analysis calculated from it does not diverge from the experimental analyses more than that calculated from the formula published by the mentioned Authors (Table V).

Table V.

| Oxides in wt. \% | I | 2 | 3 | 4 | 5 |
| :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{SiO}_{2}$ | 61.1 | 62.1 | 61.00 | 61.00 | 62.23 |
| $\mathrm{Al}_{2} \mathrm{O}_{3}$ | 0.06 |  |  |  |  |
| $\mathrm{TiO}_{2}$ | O.OI | 0.03 |  |  |  |
| FeO | <0.oI | O.I | 0.96 |  |  |
| MgO | $<0.05$ | -. 10 |  |  |  |
| CaO | 14.8 | 14.95 | 15.3 | 15.19 | 14.53 |
| BaO | 10.2 | 9.0 | $9 \cdot 3$ | 10. 38 | 9.93 |
| $\mathrm{H}_{2} \mathrm{O}$ | 13.7 | 13.7 | 13.7 | 13.42 | 13.30 |
| Total . | 100:0 | 100.0 | 100.3 | 100.00 | 100.00 |

Analyses I, 2 and 3 are published by Alfors et al. [r].
1, 2: analyses made by G. W. Putman by $d-c$ arc emission spectrograph methods;
3: wet chemical analysis made by M. Tavela;
4: chemical analysis calculated from the chemical formula given by Alfors et al. [1], $4\left[\mathrm{BaCa}_{4} \mathrm{Si}_{15} \mathrm{O}_{35} \cdot \mathrm{II} \mathrm{H}_{2} \mathrm{O}\right]$;

5: chemical analysis calculated from the chemical formula given in this work, $4\left[\mathrm{BaCa}_{4} \mathrm{H}_{2} \mathrm{Si}_{16} \mathrm{O}_{38} \cdot\right.$. $\left.\mathrm{o} \cdot 4 \mathrm{H}_{2} \mathrm{O}\right]$.

## Conclusion.

The description of the crystal structure made previously in terms of double layers of tetrahedra appears to be more suitable than that emerging from the comparison with the fibrous zeolites for a convenient introduction of macdonaldite in the Zoltai classification of silicates [II]. The crystal structure of this silicate, characterized by double layers of the apophyllite type, fills a gap in the series of structure families having as common feature
the wollastonite chain. Table VI shows the two series of structure families that can be derived respectively from the pyroxene chain and from the wollastonite chain: in the second series, the place corresponding to the "double sheets" subtype, the same as the hexagonal celsian family, was empty.

Table VI.

|  | Pyroxene chain |  | Wollastonite chain |  |
| :--- | :--- | :--- | :--- | :--- | :--- |
|  | Structure family | Repeat unit <br> or loop of <br> tetrahedra | Structure family | Repeat unit <br> or loop of <br> tetrahedra |
| Single chain . . . . . | Pyroxenes | 2 | Wollastonite | 3 |
| Double chain . . . . | Amphiboles | 6 | Xonotlite | 8 |
| Single sheet . . . . . | Micas | 6 | Apophyllite | $4-8$ |
| Double sheet. . . . . | Hexagonal celsian | $4-6$ | Macdonaldite | $4-8$ |

Table VII.

| Mineral | Cell dimensions | Chemical formula |
| :---: | :---: | :---: |
| Macdonaldite | $a=14.08 b=13.11{ }^{\text {I }} c=23.56 \AA$ | $4\left[\mathrm{BaCa}_{4} \mathrm{H}_{2} \mathrm{Si}_{16} \mathrm{O}_{38} \cdot \mathrm{IO} .4 \mathrm{H}_{2} \mathrm{O}\right]$ |
| Rhodesite |  | $4\left[\mathrm{~K}_{2} \mathrm{Na}_{2} \mathrm{Ca}_{4} \mathrm{Si}_{16} \mathrm{O}_{38} \cdot \mathrm{I} 2 \mathrm{H}_{2} \mathrm{O}\right]$ |
| Delhayelite | $b=7.04 a=13.05 c=24.65 \AA$ |  |

Probably macdonaldite is not the sole member of this structure family. 'Two minerals have crystallographic, physical and chemical properties similar to those of macdonaldite. They are rhodesite [12] and delhayelite [13] which can be compared also with fibrous zeolites. In Table VII are reported the crystallographic and chemical properties of these two silicates and of macdonaldite. The chemical formula of delhayelite is not so immediately comparable with that of macdonaldite as is that of rhodesite. The number of cations in this one is greater than that of macdonaldite, but it must be observed that the number of $\mathrm{Ca}, \mathrm{Si}$ and O , the atoms which build up the foundamental part of the structure, is the same and that in the structure of macdonaldite there is room for other cations or water molecules. However only the structural analysis could confirm or discard the hypothesis.

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